

## XRD- analysis of alginate- Impregnated Hydroxyapatite Nano-Composites

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**Abstract**—Different procedures for analysis of particle sizes by the X-ray diffraction method are compared by the example of nanoparticles of alginate-impregnated-hydroxyapatite. A modified Warren-Averbach method is proposed for the analysis of the X-ray diffraction line profile based on the approximation by the Voigt function, which yields stable solutions, and the efficiency of the method is shown. The analysis within the frame-work of the Warren-Averbach method makes it possible to restore the distribution function of nanoparticles (crystallites) over true diameters, which satisfactorily correlates with electron microscopy data. The applicability of the Warren-Averbach method to the estimation of crystallite sizes by the analysis of a single diffraction line is substantiated. The range of the applicability of the Scherrer, Williamson-Hall, Warren-Averbach, and modified Warren-Averbach methods to the substructure analysis by the X-ray diffraction is determined as depending on the method of nanostructure formation.

**Key words**—Nano particles, alginate-impregnated hydroxyapatite. X- ray diffraction.

### I. INTRODUCTION

Powder X-ray diffraction study Powder X-ray diffraction has routinely been used as a nondestructive fingerprinting technique in laboratory and industry for several decades. Every solid crystalline compound gives its own unique X-ray diffraction pattern consisting of a set of Bragg peaks. The diffraction pattern for a compound can be considered analogous to a fingerprint, or barcode, with the peak positions determined by the unit cell symmetry and lattice parameters. When we collect XRD data, we can use these fingerprints to identify not only what phases are present in our sample but also index the pattern to obtain information about the unit cell size and shape. Powder diffraction patterns of the title compound has been collected with a SIEMENS D 5000 diffractometer using CuK $\alpha$ 1 radiation ( $\lambda = 1.5406 \text{ \AA}$ ). X-ray diffraction (XRD) is an effective method for identifying the phases present in unknown polycrystalline powders. The analysis is performed by comparing the diffraction pattern collected from an unknown sample with the diffraction patterns of known compounds. The automated process is called Search/Match (S/M) analysis. XRD is an important technique in the manufacture of ceramic materials. It provides phase analysis of materials throughout the manufacturing process,. Standard  $\theta/2\theta$  data were collected using a Rigaku MiniFlex benchtop diffractometer. The experimental patterns were compared with the diffraction scans of pure compounds maintained in the ICDD Powder Diffraction File (PDF). All scans were smoothed, theta corrected, and the background was removed.

### II. MATERIALS USED

alginate-hydroxyapatite nano composites were used for XRD-analysis.

### III. RESULTS AND DISCUSSION

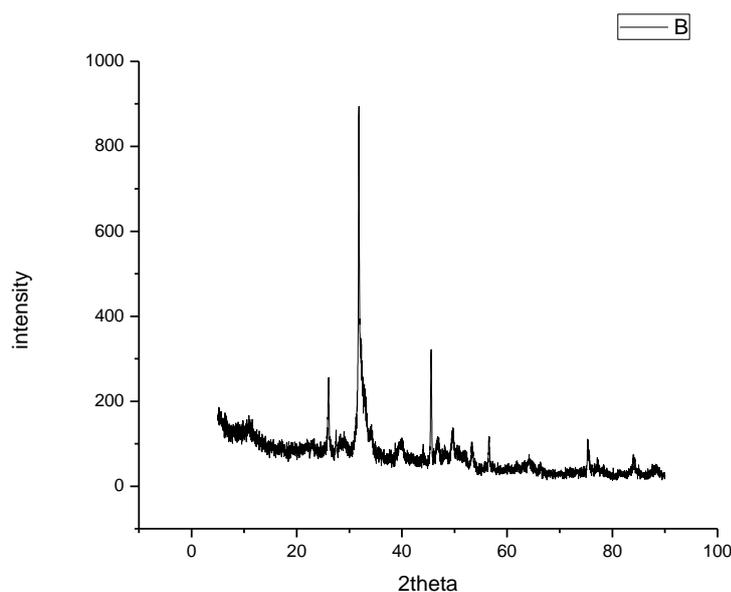


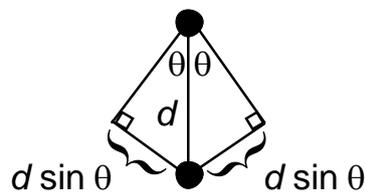
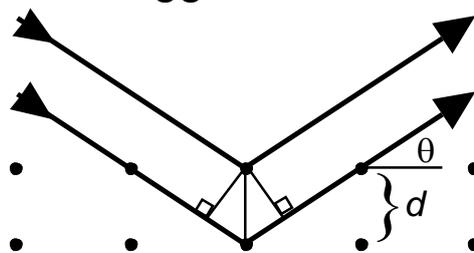
Fig1- XRD- graph of alginate-hydroxyapatite-impregnated hydroxyapatite Nano composites

#### Scherrer Equation

It is assumed that if there are N different peaks of a specific nano crystal in the range of 0 - 180° (2θ) or 0 - 90° (θ), then all of these N peaks must present identical L values for the crystal size. But, during the extensive research of the first author of this paper, on different nano ceramic crystals, which were synthesized or minerally achieved, it was surprisingly observed that each peak yields a different value and there is a systematic error on the results obtained from each peak. Further investigation approved the presence of a systematic error in Scherrer formula. In fact since  $L = \frac{K\lambda}{\beta \cos\theta}$ , if L is going to be a fixed value for  $\theta$ ,  $\cos\beta$ ,  $\lambda = L K$  different peaks of a substance, considering that K and  $\lambda$  and therefore  $K\lambda$  are fixed values, then it is essential that  $\beta \cdot \cos\theta$  be a fixed multiple during  $0 < 2\theta < 180^\circ$  or  $0 < \theta < 90^\circ$ . Suppose that for a crystallite size of 5nm, obtained at a peak of say  $2\theta = 10^\circ$  ( $\theta = 5^\circ$ ) by using  $K = 0.89$  and  $\lambda_{CuK\alpha 1} = 0.15405$  nm.  $\beta = \frac{0.89 \cdot 0.15405}{5 \cdot \cos 5^\circ} = 0.0275$  rad. Now, suppose that the N<sup>th</sup> peak of this nano crystal occurs at  $2\theta = 170^\circ$  (or  $\theta = 85^\circ$ ), then;  $\beta = \frac{0.89 \cdot 0.15405}{L \cdot \cos 85^\circ} = 0.3146$  rad. This means that the ratio of  $\frac{0.3146}{0.0275} = 11.44$ . In other words if the first  $\beta$  must be applied 170/10 for of around 2 mm on the monitor of peak has a  $\beta$  computer plot, or for example, a paper plot 21 cm width more on A4 paper, then the last peak must have a  $\beta$  170 than 22.88 mm and a base of peak more than 45.76 mm (4.576 cm). This has never been observed and cannot be true. Modified Scherrer formula is based on the face that we must decrease the errors and obtain the average value of L though all the peaks (or any number of selected peaks) by using least squares method to mathematically decrease the source of errors. We can write the basic Scherrer formula as:  $L = \frac{K\lambda}{\beta \cos\theta}$  (10). Now by making logarithm on both sides;  $\ln L = \ln K\lambda - \ln \beta \cos\theta$  (11). If we plot the results of  $\ln \beta$  against  $\ln(1/\cos\theta)$  straight line with a slope of around one and an intercept about  $\ln K\lambda$  must be obtained. Theoretically this 3. Experimental straight line must be with a slope of 45° since  $\tan 45^\circ = 1$  (Figure 1). But, since errors are associated with experimental data, the least squares method gives the best slope and most accurate  $\ln K\lambda$ . After getting the intercept, then the exponential of the intercept is obtained: Bovine bones were boiled for 2 hr to remove flesh and fat. The bones were heated at 60°C for 24 hr to remove moisture.

Crystal System	Lattice Parameter Restrictions
Cubic	$a = b = c$ $\alpha = \beta = \gamma = 90^\circ$
Tetragonal	$a = b \neq c$ $\alpha = \beta = \gamma = 90^\circ$
Orthorhombic	$a \neq b \neq c$ $\alpha = \beta = \gamma = 90^\circ$
Monoclinic	$a \neq b \neq c$ $\alpha = \gamma = 90^\circ; \beta \neq 90^\circ$
Triclinic	$a \neq b \neq c$ $\alpha \neq \beta \neq \gamma \neq 90^\circ$
Hexagonal	$a = b \neq c$ $\alpha = \beta = 90^\circ; \gamma = 120^\circ$
Trigonal*	$a = b \neq c$ $\alpha = \beta = 90^\circ; \gamma = 120^\circ$

### Bragg diffraction



For constructive interference,  
 $2(d \sin \theta) = n \lambda$

To prevent blackening with soot during heating, the bones are cut into small pieces of about 10 mm thick and heated at 400°C (bone ash) for 3 hr in air to allow for evaporation of organic substances. The resulting black bone ash was heated for 2 hr at 600°C, 700°C, 900°C and 1100°C [5]. In K L K e L (12)=  $\lambda$   $\lambda$  Having K = 0.9 and  $\lambda$  (such as  $\lambda_{Cu\alpha 1} = 0.15405$  nm), a single value of L in nanometer can be calculated

conclusion- 1) Scherrer equation systematically show increased value of nano crystalline size as d values decrease and  $2\theta$  values increase, since  $\beta \cdot \cos\theta$  cannot be maintained as constant. 2) If  $\ln\beta$  is plotted against  $\ln(1/\cos\theta)$  and least squares method is employed, the intercept gives  $\ln = K\lambda/L$ , from which to give a more accurate value of L at 600°C, 700°C, 900°C and 1100°C, the values of L for producing HA for biomaterials REFERENCES [1] P. Scherrer, "Über die Bestimmung der Inneren Struktur von Röntgenstrahlen," in the Determination of the Crystal Size of a Single Value of L can be obtained. 3) Modified Scherrer equation can provide the advantage of decreasing the errors or the curvature value of L from all or some of the different peaks. 4) In a study on natural hydroxyapatite of bovine bone, fired at 600, 700, 900 and 1100°C, 6.8, 35.5, 37.3 and 38.1 nm were respectively obtained for nano crystallite size.

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